

Magneto-Rheological Elastomers (MREs): from micro-deformation mechanisms to macroscopic instabilities and applications



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European Research Council Established by the European Commission Leiden University, Physics Department

28<sup>th</sup> May 2015



# Active materials in engineering applications



**Energy Harvesters** 





soft

#### Refreshable Braille displays



SRI (2009); Kofod et al. (2007); Carpi et al. (2010); Aschwanden & Stemmer (2006)



hard

soft







# Biological and bio-inspired active materials



#### Human Heart



Driven by electro-chemical actuation signals to create mechanical deformation





#### Human muscles







#### Magneto-active materials





Torsional suspension mounts
 (change stiffness upon application of electric current)

Vibration dampers, Haptics





[Ginder et al., 1999, Danas et al. 2012]



# Active materials in haptic interfaces



#### Tactile Teleoperation : Vibration



#### Haptic Interface : EAPs



#### Thin-metal films on PDMS



Audoly & Boudaoud, 2009; Breid and Croscby, 2009; Cai et al. JMPS, 2011

#### MRF 1-dof haptics: in-Vehicle

#### Design made by CEA LIST, France







Periquet and Lozada, 2012



#### Particle chain microstructures



#### SEM Images of 6 MREs with 11% ferromagnetic particles





# Uniform vs non-Uniform magnetic fields







Fig. 1. Non-contact mode of elongation of a magnetite loaded PDMS network due to non-uniform magnetic field Ht and Hb represents the field intensity below  $(H_b)$  and above  $(H_t)$  the magnetoelast.

**Field-Particle interactions** dominate when applied field is non-uniform.

**Particle-Particle interactions** dominate when applied field is uniform.



Fig. 2. Formation of bundles of magnetite particles in silicon oil, parallel to the field direction as seen by microscope. The concentration of magnetite in the mixture is 5 wt%. (a) No external magnetic field. (b) The magnetic induction 50 mT.

#### [Varga et al., Polymer, 2007]



# Plan of the talk



#### Fabrication and Experiments

- Particle-chain manufacturing
- Uniaxial experiments under magneto-mechanical loads.
- A micro-deformation mechanism of inter-particle motion is proposed.

#### Finite element formulations

- A unique minimum variational formulation.
- Simulations of multi-particle systems.
- Periodic unit-cells.
- Hierarchical Marginally stable MREs
  - A simple film/substrate structure.
  - Asymptotic analysis and full analytical solutions.
  - FE simulations and preliminary experiments.

# FABRICATION

Pössinger et al., SPIE Microtechnologies (2013)



# Fabrication procedure



- Particles
- Fe powder
- Average diameter size 3.5µm

- Matrix
- Soft silicone, Shore 00-20, E=0.03Mpa
- Two-part liquid RTV system





#### [Possinger et al., 2014]



# Effect of surface treatment





#### Images taken at the SEM of LMS

# EXPERIMENTS

Danas, Kankanala & Triantafyllidis, JMPS, (2012)



[Ginder et al., 1999]

MRE comprises 25% iron particles of 0.5-5μm size.

MRE cured in 0.5T magnetic field.

Curing process leads to formation of particle chains.



# Setup for magnetostriction





 Particle Chains parallel to magnetic field











# Microdeformation mechanism – Conjecture

CRUTCH

#### Parallel configuration, $\sigma/G \leq 0$



Perpendicular configuration,  $\sigma/G < 0$ 



[Klingenberg & Zukoski, 1990; Bossis & Lemaire, 1991]



# Cranton

# Parallel configuration, $\sigma/G \ge 0.075$



# **Finite Element Implementations**

Work in progress





# We do this since long time in engineering









[Danas & Aravas, 2012]





Potential energy associated with direct formulation:

$$\mathcal{P}(\mathbf{u}, \mathbf{M}, \alpha) = \int_{V} \rho_0 \left[ \Psi(\mathbf{C}, \mathbf{M}) - \mu_0 \,\mathbf{M} \cdot \hat{\mathbf{h}} - \mathbf{f} \cdot \mathbf{u} \right] \mathrm{d}V - \int_{\partial V} \mathbf{T} \cdot \mathbf{u} \,\mathrm{d}A + \int_{\mathbb{R}^3} \frac{1}{2\mu_0 J} \left| \mathbf{F} \cdot (\mathbf{\nabla} \times \boldsymbol{\alpha}) - \mu_0 \,\rho_0 \,\mathbf{M} \right|^2 \mathrm{d}V$$

ZI	$\widetilde{\mathbf{B}} = oldsymbol{ abla}  imes oldsymbol{lpha}$	$\rho_0 = \rho J :$
	$\mathbf{B} = J  \mathbf{F}^{-1}  \cdot \mathbf{b}$	${f  abla}=\partial/\partial {f X}$
	$\mathbf{B}=\widehat{\mathbf{B}}+\widetilde{\mathbf{B}}$	$J = \det \mathbf{F}$ :
<u> </u>	$\mathbf{b} = \mu_0 \left( \mathbf{h} + \rho  \mathbf{M} \right)$	$\mathbf{F} = \mathbf{I} + \mathbf{u} \boldsymbol{\nabla}$

Variation of P w.r.t the independent variables gives field equations:

$$\delta \mathcal{P}(\mathbf{u}, \mathbf{M}, \boldsymbol{\alpha}) = \mathcal{P}_{,\mathbf{u}} \, \delta \mathbf{u} + \mathcal{P}_{,\mathbf{M}} \, \delta \mathbf{M} + \mathcal{P}_{,\boldsymbol{\alpha}} \, \delta \boldsymbol{\alpha} = 0$$





	Governing equations :	Boundary/Interface conditions :		
Amper's law :	$ abla  imes \mathbf{h} = 0$	$\mathbf{n}  \times  \llbracket \mathbf{h} \rrbracket = 0$		
Zero net flux law:	$ abla \cdot \mathbf{b} = 0$	$\mathbf{n}  \cdot  \llbracket \mathbf{b} \rrbracket = 0$		
Linear momentum:	$\nabla \cdot \boldsymbol{\sigma} + \rho  \mathbf{f} = 0$	$\mathbf{n}\cdot \llbracket \sigma \rrbracket = \mathbf{t}$		
$\begin{array}{l} \hline \textbf{Constitutive equations in current configuration} \\ \textbf{Total Stress (using objectivity of $\Psi$) :} \\ \boldsymbol{\sigma} = \rho \left[ 2 \mathbf{F} \cdot \frac{\partial \Psi}{\partial \mathbf{C}} \cdot \mathbf{F}^T + \mu_0 \left( \mathbf{M} \mathbf{h} + \mathbf{h} \mathbf{M} \right) \right] + \mu_0 \left[ \mathbf{h} \mathbf{h} - \frac{1}{2} (\mathbf{h} \cdot \mathbf{h}) \mathbf{I} \right] = \boldsymbol{\sigma}^T \\ \textbf{Magnetic relations :} \qquad \mu_0 \mathbf{h} = \frac{\partial \Psi}{\partial \mathbf{M}} \qquad \mathbf{b} = \mu_0 \left( \mathbf{h} + \rho \mathbf{M} \right) \end{array}$				
$\mathbf{b}$ : magnetic field	$\mathbf{h}: \textit{h} ext{-field}$ $\mathbb{N}$	${f M}$ : spec. magnetization		
$ ho:$ mat. density $\mathbf{F}:$	deformation gradie	ent <b>f</b> : mech. body force		





Entire 2D Geometry including both air & MRE composite

2D Geometry of a representative MRE composite





### Two-particle FE analysis



2-particle chain parallel to applied magnetic field

2-particle chain perpendicular to applied magnetic field



Extension of MRE under applied magnetic field



Extension of MRE under applied magnetic field





#### Representative 2D periodic unit-cell

#### h : magnetic lines



 $\mu_p >> \mu$ 

Elastomer (non-magnetizable)

$$W(\mathbf{F}) = \frac{\mu}{2}(\overline{I}_1 - 3) + \frac{\mu'}{2}(J - 1)^2$$

Rigid Magnetizable particle (e.g., Fe)

$$W(\mathbf{F}) = \frac{\mu_p}{2} (\overline{I}_1 - 3) + \frac{\mu'_p}{2} (J - 1)^2 + c_M \mathbf{M} \cdot \mathbf{M} + f_{sat} (\mathbf{M} \cdot \mathbf{M})$$

Periodic BC's

Displacement & Stress field

$$u_i = \left(\overline{F}_{ij} - \delta_{ij}\right) X_j + u_i^*$$

$$\overline{\mathbf{T}}_{mech} = \overline{\mathbf{T}}_{tot} - \overline{\mathbf{T}}_{magn} = \overline{\mathbf{T}}_0$$

\* Magnetic potential field  $\alpha \equiv \text{periodic}$ 

(which implies that  $\alpha$  should be equal at opposite sides of the unit-cell)





- Particle shape aspect ratio :  $w = a_2/a_1$
- Particle orientation : heta

Unit-cell aspect ratio :

$$w_d = L_2/L_1$$



#### c=25%, w=0.25, wd=1





- $\mathfrak{B} = 0$  leads to no rotation but instability occurs!
- # A misorientation of the ellipse leads to change in the sign of the macroscopic magnetostriction!

# Hierarchical Marginally stable MREs

Danas & Triantafyllidis, JMPS, 2014



## Unstable behavior of MREs

# Creating

#### Structural Calculation (no periodicity)







GOAL : buckling at low magnetic fields & for a large range of layer-substrate stiffness ratios



- MRE layer / substrate block
- Magnetic field along X<sub>2</sub>
- Axial Compression along X<sub>1</sub>

- Infinite layer and substrate in X<sub>1</sub>
- Infinite substrate in X<sub>2</sub>



# Micrographs & Material selection

 $\mathbf{e}_2$ 

 $X_2$  $X_1$  $X_1$ 



MRE film Microstructure



Ginder et al. 1999, Danas et al., JMPS, 2012

Parallel configuration  $(\widehat{\mathbf{h}} \parallel \mathbf{N})$ 



Perpendicular configuration  $(\widehat{\mathbf{h}} \perp \mathbf{N})$ 



MRE comprises 25% iron particles of 0.5-5μm size.

MRE cured in 0.5T magnetic field.

Curing process leads to formation of particle chains.

# An asymptotic analysis



#### Energetics of the MRE film

$$\psi_l(\mathbf{F}, \mathbf{M}) = G_l \left[ \frac{H^2}{12(1-\nu_l)} (w_{,11})^2 + \frac{C_7}{2M_s^2} (\lambda_2 M)^2 \right]$$

$$w(X_1) \equiv u_2(X_1, \frac{H}{2}), \ M(X_1) \equiv M_2(X_1, \frac{H}{2})$$



$$\psi_s = \frac{G_s \gamma}{2(1-\nu_s)} w^2$$

Using small strain kinematics :

$$\lambda_2 = 1 + \epsilon_{22}, \ J = 1 + \epsilon_{11} + \epsilon_{22}; \quad \epsilon_{22} = -\frac{\nu_l}{1 - \nu_l} \epsilon_{11}, \ \epsilon_{11} = \frac{1}{2} (w_{,1})^2$$







The potential energy of the film/substrate system becomes:

$$\mathcal{P} = \int_0^L \left\{ \frac{G_l H^3}{12(1-\nu_l)} (w_{,11})^2 - \frac{\mu_0 H(\hat{h})^2}{4(1+\beta)^2} \frac{1-2\nu_l+2\beta\nu_l}{1-\nu_l} (w_{,1})^2 + \frac{\omega G_s \gamma}{2(1-\nu_s)} w^2 \right\} \mathrm{d}X_1$$

The second variation of the potential energy then leads to (Biot, 1937):

$$(\mathcal{P}_{,ww}\Delta w)\delta w = \frac{G_l H^3}{6(1-\nu_l)}\Delta w_{,1111} + \frac{\mu_0 H(\hat{h})^2}{2(1+\beta)^2} \frac{1-2\nu_l+2\beta\nu_l}{1-\nu_l}\Delta w_{,11} + \frac{\omega G_s\gamma}{1-\nu_s}\Delta w = 0$$

Substitution of a sinusoidal mode on w, i.e.,  $\Delta w = W \sin(\omega X_1)$ 

$$\hat{h}_c \sim (G_s/G_l)^{1/3}$$
;  $(\omega H)_c \sim (G_s/G_l)^{1/3}$ 





Second variation of the potential energy:

$$(\mathcal{P}_{,\mathbf{gg}}(\mathbf{g}_0(\mathbf{\Lambda}_c),\mathbf{\Lambda}_c)\Delta\mathbf{g})\delta\mathbf{g} = 0, \qquad \mathbf{\Lambda}_c = \{\lambda_1^c, \hat{h}_c\}$$

• Look for solution of the type :

$$\Delta u_j(X_1, X_2) = \exp\left(i\omega X_1\right) \left[\sum_I \Delta U_j^I \exp\left(\zeta_I \omega X_2\right)\right],$$

$$\Delta \alpha(X_1, X_2) = \exp\left(i\omega X_1\right) \left[\sum_I \Delta \mathcal{A}^I \exp\left(\zeta_I \omega X_2\right)\right],$$

 $\zeta_{I}\;$  : complex roots obtained by the characteristic equation resulting from the Euler-Lagrange equations

# Results



Purely Magnetic Loading and Zero Stretching

#### Purely Mechanical Loading







- **#** The instability is a combination of a structural and microstructural instability.
- Hierarchical instabilities explain the difference between Par/Perp
- **%** More studies need to be done at the different scales







#### Danas & Triantafyllidis, JMPS, 2014

CINIC



Shear Modulus of Substrate:  $G_s$ 

Shear Modulus of Layer:  $G_l$ 

# Post-bifurcation with Finite Element

Work in progress





# MRE layer / substrate simulation

















## No prestretch : critical field and modes







## Prestretch, critical field and modes





Increasing stretch ratio  $\lambda$ 

How With increasing substrate stiffness & stretch and thickness we form creases...

ℜ The interplay of mechanical and magnetic fields is very rich...



# Fabrication and Experiments underway...



MRE film

#### The different scales...

# Imm Imm

10mm

50µm







# THANK YOU FOR YOUR ATTENTION



Financial Acknowledgement: ERC-Starting-Grant: MAGNETO



European Research Council Established by the European Commission

- Ecole Polytechnique
  - http://www.polytechnique.edu
- Laboratory of Solid Mechanics
  - <u>http://www.lms.polytechnique.fr</u>
- Kostas Danas webpage
  - http://hera.polytechnique.fr/users/kdanas



20



Post-bifurcation response : Ridge or Crease

CIERCIS





Shear Modulus of Substrate:  $G_s$ 

Shear Modulus of Layer:  $G_l$ 



# MRE Micrographs & Material Fabrication



Microstructure



[Ginder et al., 1999]

MRE comprises 25% iron particles of 0.5-5μm size.

MRE cured in 0.5T magnetic field.

Curing process leads to formation of particle chains.



- Combine mechanical and magnetic loads in such a way that:
  - Using first purely mechanical loads bring the system near (but not at) a critically stable regime
  - Trigger the instability with a tiny magnetic field



# Effect of surface treatment





#### Images taken at the SEM of LMS



#### c=25%, w=wd=0.25





 $\hat{h}/\rho_0 M_s$ 

- $\mathfrak{H}$  $\theta$ =0 leads to no rotation but instability occurs!
- H A misorientation of the ellipse leads to change in the sign of the macroscopic magnetostriction!



# Wrinkling & Creasing with EAPs





- Particle vol % : 5-35%
- Number of particles : 1







- **#** Monotonic dependence on vol% for small fields
- % Non-monotonic dependence on vol% for large fields due to magneto-mechanical interplay at finite strains



Interface/Boundary conditions at :

$$\underbrace{ \text{Essential}}_{\{\Delta\alpha\}_l = \{\Delta\alpha\}_a} \begin{pmatrix} X_1, X_2 \end{pmatrix} \in \mathbb{R} \times \{H\} & \underline{\text{Natural}}\\ \{L^{uu}_{i2kl} \Delta u_{k,l} + L^{u\alpha}_{i2k} \Delta \alpha_{,k}\}_l = 0\\ \{L^{\alpha u}_{2jk} \Delta u_{j,k} + L^{\alpha \alpha}_{2j} \Delta \alpha_{,j}\}_l = 0 \end{cases}$$

NOTE: Incremental moduli:  $L_{ijkl}^{uu} = \frac{\partial^2 \mathcal{P}}{\partial F_{ij} \partial F_{kl}}, L_{ijk}^{u\alpha} = \frac{\partial^2 \mathcal{P}}{\partial F_{ij} \partial \widetilde{A}_k}, L_{ij}^{\alpha\alpha} = \frac{\partial^2 \mathcal{P}}{\partial \widetilde{A}_i \partial \widetilde{A}_j}$ 



The use of Euler-Lagrange equations together with the boundary conditions lead to 10x10 system:



$$\sum_{q=1}^{10} \mathcal{D}_{pq}(\mathbf{\Lambda}, \omega H) \mathcal{H}_q = 0, \quad 1 \le p \le 10$$

where

$$\mathcal{H}_q = \mathcal{H}_q(\Delta U^I, \Delta \mathcal{A}^I)$$



$$\det[\mathcal{D}(\mathbf{\Lambda},\omega H)] = 0, \qquad \mathbf{\Lambda}_c = \{\lambda_1^c, \hat{h}_c\} = \inf_{\omega H} \mathbf{\Lambda}(\omega H)$$

#### **#** Interesting to see what Catastrophe Theory can tell us?