

Modeling of porous materials consisting of isotropic and anisotropic matrix and implications on deformation localization



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# Cyclic loading of a porous unit-cell



#### Void shapes as a result of fabrication



#### Aluminium Alloy, A319, Lost Foam Casting

#### $D\times H=4.8\mathrm{mm}\times 3.9\mathrm{mm}$





#### [Limodin et al., 2013]

3-D close up of coalescing elongated voids in the L-S plane in one column

Courtesy of Limodin and Charkaluk

#### The geometry of the unit-cell



#### Periodic boundary conditions

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 $\sigma \cdot n$  anti – periodique,  $v = D \cdot x + v^*$ ,  $v^*$  periodique



#### Cyclic loading at constant Xs and Lode







#### FEM simulations: Qualitative results





















Effect of kinematic hardening



$$\dot{X}^{(1)} = q_h C_1 D_M^p, \quad \dot{X}^{(2)} = q_h C_2 D_M^p - \gamma_2 \Delta \epsilon^p X^{(2)}$$

$$\longrightarrow X_{\Sigma} = 3, \ \theta = 0$$

 $N_r = 40, q_h = 0$ 













## Switch to

# Monotonic loadings



#### Creep tests in single crystals



#### Nearly spherical pores



#### Irregular void shapes



After 78h

#### h=51.8mm specimen







[Shrivastava, Thesis, 2013]



#### **Dual Phase Thin Sheets**



1) Large Thin sheets, Active zone 4mm, Grain size  ${\sim}20\mu\text{m}$ 



2) Small Thin sheets, Active zone 200µm, Grain size ~20µm







#### Dual Phase Thin sheets inside SEM









Clanter

Joint work with Eva Heripre, LMS, SEM



#### In situ SEM Dual Phase experiments







 $\varepsilon_{11} + \varepsilon_{22}$ 









#### Dual Phase Thin sheets : Small specimens



CORPORATION

ferrite (dark color) martensite (light color) mm 0.15 mm Transverse direction Plate thickness =0.08mm R0.2 R0.4 Thickness direction 2.2 1.00.22.9**NIPPON STEEL &** 4.9 SUMITOMO METAL













#### Material & Structural failure





# Linear Comparison Composite (LCC)

&

# Structure of equations in a nutshell



#### Evolution of microstructure



Kailasam & PC, JMPS, (1998), Danas & PC, Eur. J. Mech A (2009), Danas & Aravas, Comp. Part B (2012)

#### The variational problem

 $\mathbf{n}^{(3)}$ 

 $a_3$ 

 $a_2$ 

 $\mathbf{n}^{(2)}$ 

•  $\mathbf{e}^{(3)}$ 

 $\mathbf{k}\mathbf{e}^{(1)}$ 

 $\mathbf{n}^{(1)}$ 

 $\mathbf{e}^{(2)}$ 

 $a_1$ 



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$$\widetilde{U}(\overline{\boldsymbol{\sigma}}) = (1 - f) \min_{\boldsymbol{\sigma} \in \mathcal{S}(\overline{\boldsymbol{\sigma}})} \int_{V_m} U(\boldsymbol{\sigma}, \mathbf{x}) \mathrm{d}V$$

- Requires choice of divergence free stress fields
- Equivalent formulation in strains requires compatible strain fields
- The modified variational principle (Ponte Castaneda, JMPS, 1991)

$$\widetilde{U}_{VAR}\left(\overline{\boldsymbol{\sigma}}\right) = \sup_{\mathbf{M}} \left\{ \widetilde{U}_{L}\left(\overline{\boldsymbol{\sigma}};\mathbf{M}\right) + (1-f) \min_{\hat{\boldsymbol{\sigma}}} \left[U\left(\hat{\boldsymbol{\sigma}}\right) - U_{L}\left(\hat{\boldsymbol{\sigma}},\mathbf{M}\right)\right] \right\}$$

 $M: \ \mbox{ Is the elastic compliance of the LCC }$ 

- The "min" operation is relaxed
- The M is chosen so that simple estimates are derived



#### **Evolution of microstructure**

- Porosity evolution:  $\dot{f} = (1-f)\dot{arepsilon}_{kk}^p$
- Void shape evolution:  $\dot{w}_i = \alpha_w \, w_i \left( \mathbf{n}^{(3)} \mathbf{n}^{(3)} \mathbf{n}^{(i)} \mathbf{n}^{(i)} \right) : \dot{\boldsymbol{\varepsilon}}^v, \ i = 1, 2$
- Void orientation evolution:

$$\dot{\mathbf{n}}^{(i)} = oldsymbol{\omega}(\mathbf{\Omega}, \dot{oldsymbol{arepsilon}}^v) \cdot \mathbf{n}^{(i)}$$

## Crystal Plasticity Yield functions

# Modified VAR (MVAR)

(Mbiakop, Constantinescu, Danas, IJSS, in press)

#### The constitutive response of the matrix

CITATION



$$U(\boldsymbol{\sigma}) = \sum_{s=1}^{K} \Psi^{(s)}(\tau^{(s)}) = \sum_{s=1}^{K} \frac{\dot{\gamma}_0^{(s)} \tau_0^{(s)}}{n+1} \left(\frac{|\tau^{(s)}|}{\tau_0^{(s)}}\right)^{n+1}$$

Schmid tensor:  $\mu^{(s)} = \frac{1}{2} \left( \mathbf{m}^{(s)} \otimes \mathbf{s}^{(s)} + \mathbf{s}^{(s)} \otimes \mathbf{m}^{(s)} \right)$ CRSS stress & slip:  $\tau^{(s)} = \boldsymbol{\sigma} \cdot \boldsymbol{\mu}^{(s)} \quad \gamma^{(s)} = \frac{\partial \Psi^{(s)}}{\partial \tau^{(s)}} (\tau^{(s)})$ 





For a J2-plastic matrix (PC, 1991)

$$\widetilde{\Phi}_{mvar}(\overline{\boldsymbol{\sigma}}) = \max_{s=1,K} \left\{ \sqrt{\frac{\overline{\boldsymbol{\sigma}} \cdot \mathbf{m}^{(s)} \cdot \overline{\boldsymbol{\sigma}}}{1-f}} - (\tau_0)^{(s)} \right\} = 0$$
$$\mathbf{m}^{var,(s)} = \boldsymbol{\mu}^{(s)} \otimes \boldsymbol{\mu}^{(s)} + \frac{f}{K} \left( \mathbf{Q}^{-1} - \sum_{s=1}^{K} \boldsymbol{\mu}^{(s)} \otimes \boldsymbol{\mu}^{(s)} \right), \quad \mathbf{Q} = g(\mathbf{P}, \boldsymbol{\mu}, \boldsymbol{\mathcal{J}})$$

A simple correction on the hydrostatic part only

$$\mathbf{m}^{(s)} = \mathbf{m}^{var,(s)} + (q_J^2 - 1) \mathcal{J} \cdot \mathbf{m}^{var,(s)} \cdot \mathcal{J}$$
$$q_J = \sqrt{\frac{5}{2}} \frac{1 - f}{\sqrt{f} \ln(1/f)}$$

Mbiakop Constantinescu and Danas, IJSS, in press 2D; and in preparation 3D



All models show the same features and minor differences are observed

Han et al, IJSS, 2013; Paux et al., EJMSOL/A, 2015; Mbiakop et al., in prep.



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#### Deviatoric planes for spherical pores







Increasing superposed positive pressure



#### Effect of void shape in FCC

frontières





#### Effect of void shape in FCC : dev plane







#### FEM contours : f=1%



#### FCC

#### Total slip $\gamma_{tot}/\gamma_0$





BCC





HCP





#### 2D cross-sections : FCC crystal



f=1%









# Isotropic MVAR ABAQUS UMAT

# Qualitative comparisons with

# experiments

Danas & Aravas (2012), Composites Part B

But see also

(Son, Maziere, Danas, Besson, IJSS, soon in press) (GVAR)





For a J2-plastic matrix (PC, 1991)

$$\Phi(\boldsymbol{\sigma}) = \widehat{\sigma}_e - \sigma_y(\varepsilon_M^p) = 0, \quad \widehat{\sigma}_e = \sqrt{\frac{\sigma_{ij} m_{ijkl} \sigma_{kl}}{1 - f}}$$

$$\mathbf{m}^{var} = \frac{3}{2} \mathcal{K} + \frac{3f}{1-f} \mathbf{Q}^{-1}, \quad \mathbf{Q} = g(\mathbf{P}, \mathcal{K}, \mathcal{J})$$

 A simple correction on the hydrostatic part only (PC, 1992; Michel & Suquet, 1992)

$$\mathbf{m}^{mvar} = \mathbf{m}^{var} + (q_J^2 - 1) \, \mathcal{J} \cdot \mathbf{m}^{var} \cdot \mathcal{J}, \qquad q_J = \frac{1 - f}{\sqrt{f} \ln(1/f)}$$

The predictions for arbitrary void shapes are extrapolations of this approximation



#### Yield Surfaces: Isotropic microstructures





MVAR is symmetric w.r.t.  $\sigma_m$  and slightly more compliant than SOM

But see bounds of PC, 2013, Agoras and PC, IJSS, 2014



#### Yield Surfaces: Void shape effects



MVAR is identical to the SOM in the purely hydrostatic limit for any shape



#### **Evolution of microstructure**

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- Void orientation evolution:

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#### Courtesy of Mathieu Dunand and Dr. Mohr



#### Macroscopic force-displacement curves



**Shear-dominated loading** 25 20  $F_{x}$  (kN) 15  $u_x = 10 u_y$ 10 MVAR — 5 GUR ---- $f_0 = 1\%$ MISES ----0.5 1.5 2 2.5 3 3.5 0 1 1  $u_x \text{ (mm)}$ 

#### **Tension-dominated loading**











#### Position of fracture initialization







# Qualitative comparison with experiments:

Arcan Specimen



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#### Ghahremanineshad and Ravi-Chandar, IJF, 2013



#### Geometry of Arcan Specimen













- Fracture occurs above the initial notch
- Very high strains to fracture
- Fracture very localized

#### Ghahremanineshad and Ravi-Chandar, IJF, 2013



#### Geometry of Arcan Specimen















- Fracture occurs above the initial notch
- Very high strains to fracture
- Fracture very localized

#### Ghahremanineshad and Ravi-Chandar, IJF, 2013



Unpublished Results

# Qualitative comparison with experiments:

# Shear of a Notched Cylindrical

Specimen



#### Large shears : no failure predicted



 $W_1$ 

S, Mises (Avg: 100%) 1.85 1.71 1.56 1.41 1.26 1.41 1.26 0.96 0.81 0.66 0.51 0.36 0.21 0.06



No failure predicted. Need for additional mechanisms (e.g., damage nucleation, crack nucleation)

# Hole expansion test:

## Effect of initial void shape due to

Forming process



#### Bore expansion : Geometry and ICs



**Unpublished Results** 



#### FEM simulations: Qualitative results







**Unpublished Results** 

#### Some concluding remarks



#### What we see in experiments

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- Fracture and localization is a multi-scale problem
- At the small scales deformation localization and cracking can co-exist affecting one each other in a non-trivial manner.
- Pre-existing voids or nucleated ones are a reality?

#### Porous material modeling

- Void shapes and orientations are critical in porous plasticity
- In crystal plasticity voids lead to non-trivial effects (HCP-basal incompressible)
- Localization of strains is non-trivial in a random distribution of porosity

#### Geometry, loading and materials

- Geometry, loading and materials are coupled!
- A sufficiently good plasticity model could tell you if pores are the source of failure or not!
- Stress fields depend on the use of the model (check Strain triaxiality)



#### **THANK YOU FOR YOUR ATTENTION**



- Laboratory of Solid Mechanics (LMS)
  - http://www.lms.polytechnique.fr

- Kostas Danas webpage
  - http://hera.polytechnique.fr/users/kdanas

- **Ecole Polytechnique** 
  - http://www.polytechnique.edu



- National Center of Scientific Research (C.N.R.S) webpage
  - http://www.cnrs.fr/index.php





#### Modified VAR model

- Simple modification of the earlier variational method.
- Easily implemented in FEM and very stable numerically
- Time of calculations in the order of Gurson model
- Includes shape and rotation effects ab initio
- x Does not have the character of a bound any more

#### MVAR & Experimental geometries

- Even for a low triaxiality macroscopic load the mode of structural failure is driven by geometrical aspects leading to high triaxiality locally and void growth.
- MVAR and GUR exhibit significant quantitative differences at the local level.
- Qualitatively, MVAR tends to localize more abruptly than GUR (similar to the experiments), but differences are very small until that localization point.
- The evolution of the void shape leads in general to more criticalbehavior, which in turn leads to faster material & geometrical failure locally.



Ponte Castañeda (2002), Aravas & Ponte Castañeda (2004), Danas & Ponte Castañeda (2009)



#### Effect of void shape in FCC

frontières





#### Void shapes in HDPE



# 30 µm (a)

#### voids elongated in the loading direction





#### isotropic voids





(b)



#### Evolution of voids: ductile crack opening

#### Al-Cu-Mg, AA2139 alloy





- 3-D void distribution and morphology in the L-T plane
- Volume of 350x350x180μm
- Void content appears higher due to projection

 3-D close up of coalescing elongated voids in the L-S plane in one column

10

 Significant void shape evolution



# Tresca Yield functions (Bonus CP)

# Modified VAR (MVAR)

(Mbiakop, Constantinescu, Danas, in preparation)



#### Spherical voids: Effect of Lode angle







#### Void shape effect









#### Elastic material properties

$$E = 1000 \sigma_0, \ \nu = 0.3, \ \varepsilon_0 = \sigma_0/E$$

Hardening exponent *N* 



Initial values for the microstructural variables (initially isotropic)

$$\varepsilon^p = 0, \ f_0 = 1\%, \ w_1 = w_2 = 1$$
 at time  $t = 0$ 





Stress triaxiality: 
$$X_s = \sigma_m / \sigma_e$$
  $\sigma_m = \sigma_{kk} / 3, \ \sigma_e = \sqrt{3 \sigma'_{ij} \sigma'_{ij} / 2}$ 

\* Lode angle : 
$$L = -\cos 3\theta = -\frac{27}{2} \det \left(\frac{\sigma'_{ij}}{\sigma_e}\right) \quad \begin{bmatrix} 0 \le \theta \le \frac{\pi}{3} \\ -1 \le L \le 1 \end{bmatrix}$$

IMPORTANT NOTE: In the presented models (MVAR, VAR, SOM) the response is anisotropic and depends on the entire stress tensor in general loadings